

Synthesis of CNT film on the surface of micro-pyramid array and its intense pulsed emission characteristics

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We developed a new scheme to suppress the electric-field-screening effect in high growth density of a carbon nanotube (CNT) film during its intense pulsed emission. We synthesize the CNT film on a tridimensional surface (t-CNT film). The tridimensional surface includes wet etched silicon pyramids, and the Ni layer is electroless plated thereon. The intense pulsed emission characteristics of the t-CNT and planar-grown CNT (p-CNT) films were measured using a diode structure in single-pulse mode. The even turn-on field decreased from 5.5 V/ μm for p-CNTs to 2.8 V/ μm for t-CNTs, and the peak emission current increased from 232 A for p-CNTs to 324 A for t-CNTs at a peak field intensity ~ 12.2 V/ μm . The peak current of the t-CNT film increased by $\sim 39.7\%$ over the p-CNT film. It is clear that the micro-pyramid array can effectively suppress the field screening effect to improve the electron-emission of CNT films.

carbon nanotubes, micro-pyramid array, field-screening effect, intense pulsed emission, emission current

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Carbon nanotubes (CNTs) have attracted considerable attention for their potential field-emission applications because of their excellent properties [1–3]. These properties make CNTs a good candidate for cold cathodes in field-emission displays, high-resolution electron-beam instruments, lamps, X-ray sources, and microwave amplifier tubes [4–9]. Improving current emission is very important for using CNTs as electron sources. However, when the density of CNTs increases, the field emission of a CNT cathode will be depressed by the field screening effect [10,11], resulting in a higher turn-on voltage and lower emission current. Therefore, it is necessary to find a way to suppress the field-screening effect.

In this paper, we report the synthesis of a CNT film on the surface of a micro-pyramid array and its intense pulsed

emission characteristics (the as-grown CNT film is hereafter called the “t-CNT film”). The function of the micro-pyramid array in suppressing the field screening effect can be easily found by comparing the emission characteristics of the t-CNT film with that of the planar-grown CNT (p-CNT) film.

1 Experimental

1.1 Sample preparation

(1) Materials and reagents. An n-type silicon (Si; Tianjin Institute of Semiconductor Technology, Tianjin, China) wafer (100) with a resistivity of 10^{-2} – 10^{-3} $\Omega\text{ cm}$ and 2-in diameter was used as the substrate material. We used a negative photoresistor for photolithography to transfer the designed pattern to the silicon dioxide layer. Buffered oxide

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etchants (BOE, home-made) were used for etching the silicon dioxide windows. Potassium hydroxide (KOH; Tianjin Kaitong Chemical Reagent Co. Ltd, Tianjin, China) solution with a concentration of 40 wt% was used as an etchant in the anisotropic wet etching to form Si pyramids. The electroless nickel plating solution was composed of nickel sulfate (Tianjin Fengchuan Chemical Reagent Technological Co. Ltd, Tianjin, China), sodium hypophosphite (NaH_2PO_2 ; Tianjin Kermel Chemical Reagent Co. Ltd, Tianjin, China), and trisodium citrate (Tianjin Kermel Chemical Reagent Co. Ltd). Palladium (II) chloride (Tianjin Kermel Chemical Reagent Co. Ltd) was used as the catalyst for the Ni plating. Iron phthalocyanine (FePc; Tokyo Chemical Reagent Co. Ltd, Tokyo, Japan) was used for growing the CNT film.

(2) Preparation procedure. Si pyramids were aligned on the Si surface by wet etching to form a tridimensional surface. The preparation of the Si pyramids was as follows. The silicon wafer was oxidized to form a silicon dioxide layer with a thickness of more than 300 nm. We did photolithography for the resistor to transfer the designed pattern onto the silicon dioxide layer. The mask for the photolithography was made on a plastic film by a photoplotter (JOINYOU 2032A, Shenzhen, China). We etched the silicon dioxide layer to form etching windows. The etching temperature in the BOE etchants was $\sim 40^\circ\text{C}$. We formed Si pyramids by anisotropic wet etching. The etching temperature of the KOH solution was $90\text{--}100^\circ\text{C}$. When the Si pyramids formed, the Si wafer was immersed in de-ionized water and then in the BOE etchants to remove residual silicon dioxide layer.

A nickel film with $\sim 1\ \mu\text{m}$ thickness was electroless plated on the tridimensional surface. Before the plating of the Ni layer, the Si wafer was immersed in palladium (II) chloride aqueous solution for 30 s. Then the catalyzed Si wafer was put into the plating solution for 5 min at 60°C . The pH value of the solution was adjusted to be 8–10 by adding ammonia to the solution.

The CNT film was synthesized by pyrolysis of FePc [12]. The temperature for CNT growth was $\sim 900^\circ\text{C}$. The growth time was ~ 10 min.

1.2 Study on the morphologies of samples

A scanning electron microscope (SEM, JEOL JSM-6700F) was used to study the morphologies of the substrate and CNT film. The morphology of the tridimensional surface (Si pyramids) was observed with an optical microscope (OM).

1.3 Measurement of intense pulsed emission characteristics

We measured intense pulsed emission characteristics using a diode structure with a pulse-forming network under a vacuum of $\sim 5 \times 10^{-4}$ Pa. The anode-cathode distance was

~ 14 cm. The half-value width of the pulsed voltage was ~ 100 ns. We measured the t-CNTs and p-CNTs under the same pulsed electric field to compare their emission characteristics.

2 Results and discussion

2.1 Morphologies of t-CNTs

Figure 1 shows the OM image of the pyramids. The distance between nearest-neighbor pyramids is $100\ \mu\text{m}$, and the length of bottom margin is $\sim 30\ \mu\text{m}$.

KOH is by far the most common anisotropic wet etchant. Its etch rates for single-crystal silicon depend on the crystallographic direction. For example, $\{111\}$ planes are etched about 100 times more slowly than are $\{100\}$ planes [13]. Therefore, the four oblique planes of a pyramid are in fact four $\{111\}$ planes, making a 54.7° angle with respect to the horizontal (100) surface. For the $\sim 30\text{-}\mu\text{m}$ bottom margin, we calculate the height of a pyramid to be $\sim 21.2\ \mu\text{m}$.

Because the mask used for lithography was made on a plastic film by a photoplotter, the shapes and precision of the designed square-dot array are not good, resulting in a pyramid array that is not very regular (Figure 1).

Figure 2 shows the SEM images of the t-CNTs. Figure 2(b) and (c) is the magnified images of the CNTs on a pyramid in (a). CNTs on the bottom flat are shown in Figure 2(d). We see from Figure 2 that the sample includes two kinds of morphologies: one is in the pyramid areas and the other the planar area. CNTs in the pyramid areas grew on the surface of the micro-pyramid array homogeneously, with left tussock-like CNTs with $2\ \mu\text{m}$ length and $\sim 20\text{--}50$ nm diameters on the tips of the pyramids. The tussock-like CNTs are slanted with the pyramid surfaces, perpendicular to their substrates. The diameters of CNTs in the planar area are in the same range as those in pyramid areas. The range of CNT diameters in this study is in agreement with those in published reports [12,14]. Like those in references [12,14],

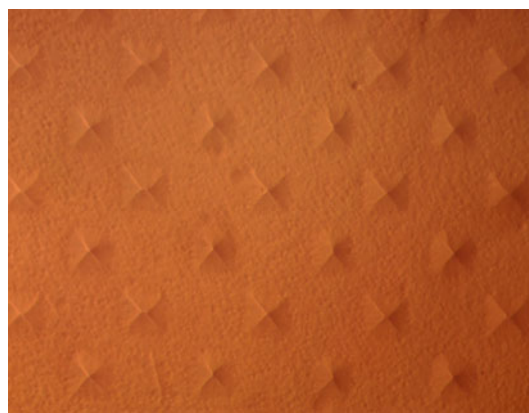


Figure 1 OM image of micro-pyramid array. The distance of nearest neighbor pyramids is $100\ \mu\text{m}$.

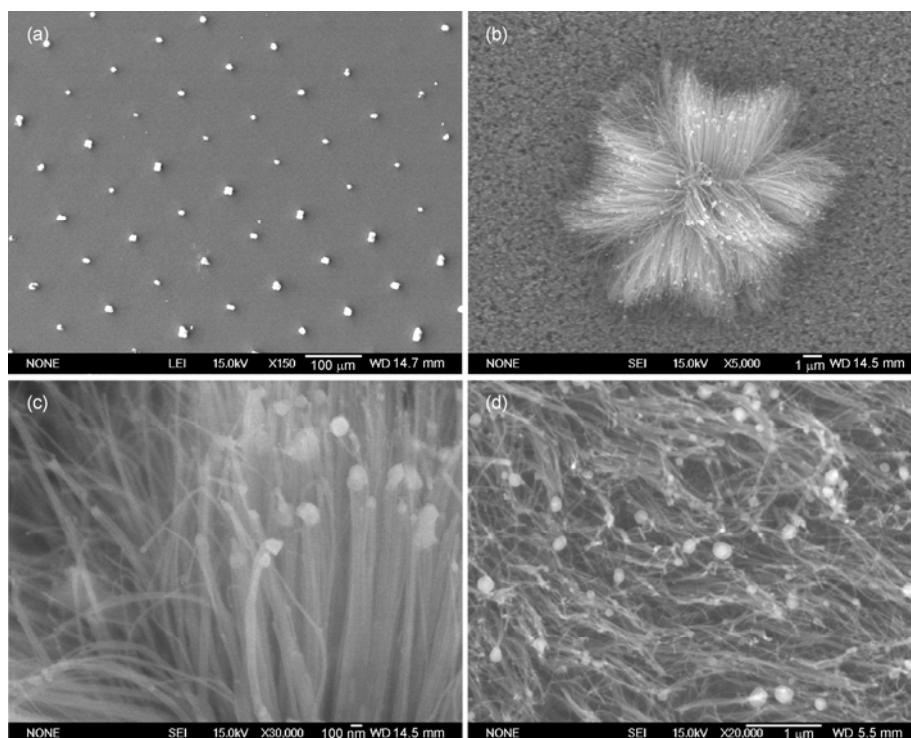


Figure 2 SEM images of CNTs grown on a tridimensional surface. (a) Total view; (b) CNTs grown on a pyramid tip; (c) magnified image of (b); (d) CNTs grown on the bottom flat.

the as-grown t-CNTs and p-CNTs are very dense, but there are some gaps in the t-CNTs due to the orientation-dependent CNT growth.

Our method of growing CNTs is the same as that in references [12,14], in which the results of different authors show good consistency. The reader who is interested in the details of CNTs prepared by the pyrolysis of FePc can see these references.

2.2 Characteristics of intense pulsed emission in single-pulse mode

Figure 3 shows the wave curves of the pulsed current and voltage vs. time, where (a) and (b) correspond to t-CNTs and p-CNTs and curves C_1 and C_2 correspond to the electric field and current, respectively. We see that the current curve agrees in general with the voltage. The peak of the displacement current, which is caused by the capacitor charge and discharge effects, cannot be found easily in Figure 3. Therefore, the capacitor charge and discharge effects can be neglected. The currents of C_2 in Figure 3(a) and (b) are mainly from electron emission. For a peak field intensity of ~ 12.2 V/ μm , the peak emission currents and equivalent current densities are 324 A and 16.5 A/ cm^2 for t-CNTs and 232 A and 12.1 A/ cm^2 for p-CNTs. The emission current increases by $\sim 39.7\%$ from the p-CNT to the t-CNT film. Points A and B are the emission turn-on and turn-off point; the turn-on and turn-off electric field can be deduced from A' and B'. For the t-CNT film, the values of the turn-on and

turn-off electric field and their mean value are 3.1, 2.7 and 2.8 V/ μm , respectively. Those for the p-CNT film are 5.0, 6.0 and 5.5 V/ μm . Table 1 compares the intense pulsed emission characteristics of t-CNTs and p-CNTs.

2.3 Mechanism of improving emission characteristics

The improvement of emission characteristics for the micro-pyramid-array surface can be understood from two aspects. On one hand, the tridimensional surface increases the growth area of the CNT film, making more CNTs participate in electron emission. On the other hand, the micro-pyramid array suppresses the field-screening effect, resulting in a stronger electric field on the CNT emitters, which increases the emission current and decreases the turn-on electric field. Both aspects increase the emission of the t-CNT film, resulting in a stronger electron emission.

Although the growth area of the CNT film increases because of the tridimensional surface, the area of the tridimensional surface increases by only $\sim 15\%$ over the planar surface in our case. Comparing with the $\sim 39.7\%$ increase of the emission-current ratio, we see that more current increase comes from suppressing the field-screening effect.

However, the increase of $\sim 24.7\%$ in emission current ratio, which is caused by suppressing the electric-field-screening effect, is not very distinct. This is because the quantity of pyramids per unit area is small. The basal area of the pyramids occupies only a small percentage, less than 10%, of the total area of the planar substrate. In fact, most of the area of our

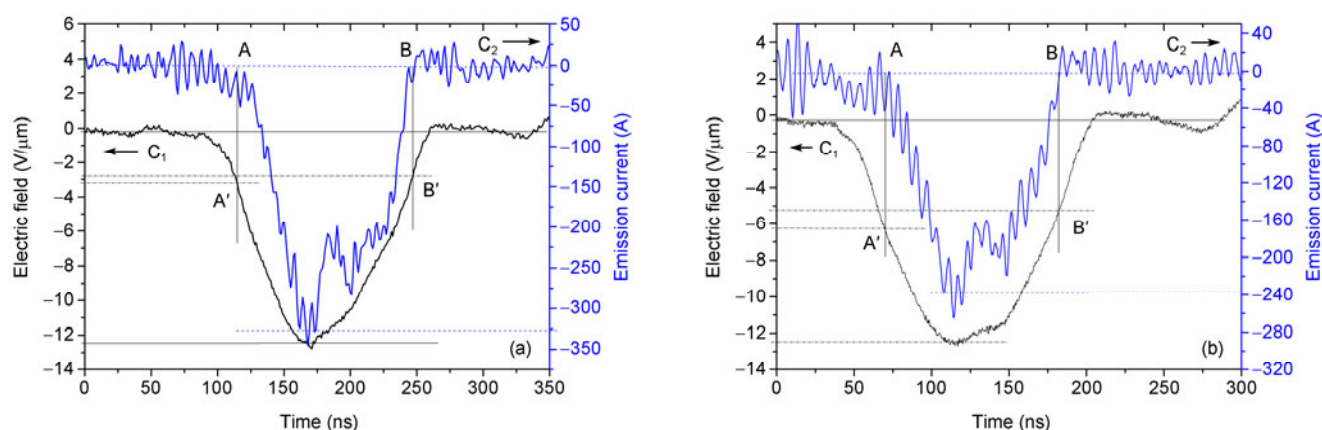


Figure 3 Waveforms of pulsed current and voltage corresponding to t-CNTs (a) and p-CNTs (b), in which C₁ and C₂ are electric field and current curves, respectively.

Table 1 Comparison between CNTs grown on tridimensional and planar surfaces

Sample type	Turn-on field (V/μm)	Emission current (A)	Current density (A/cm ²)
t-CNTs	2.8	324	16.5
p-CNTs	5.5	238	12.1

t-CNT sample consists of planar-grown CNTs. The emission current of the t-CNT sample is the total of those from the pyramid and planar areas.

We think that the emission of a t-CNT film can be further improved by optimizing the size and density of pyramids on the substrate in designing the tridimensional surface. For example, we can increase the quantity of pyramids in a given area, control the size of pyramids according to their electric field distribution, and optimize them to a proper status to obtain a maximized emission current. Further research is going on, and the results will be reported later.

3 Conclusion

We presented a new scheme to suppress the field-screening effect by growing CNTs on a tridimensional surface. The intense pulsed emission characteristics of t-CNT and p-CNT films were measured using a diode structure in single-pulse mode. The even turn-on field decreased from 5.5 V/μm for p-CNTs to 2.8 V/μm for t-CNTs, and the peak emission current increased from 232 A for p-CNTs to 324 A for t-CNTs at a peak field intensity of ~12.2 V/μm. The peak current of t-CNTs increased by ~39.7% over the p-CNTs.

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